# Lecture 2: Molecular Design and Synthesis of Biomaterials I: Biodegradable Solid Polymeric Materials (continued)

Last time:	chemistry and physical chemistry of degrading polymeric solids for biomaterials			
Today:       Theory of polymer erosion         Enzymatic degradation of synthetic biomaterials       Designing degradable materials				
Reading:		A. Gopferich, "Mechanisms of polymer degradation and erosion,' <i>Biomaterials</i> <b>17</b> , 103 (1996) Ratner p. 243-259		
Supplementary	y Reading:	R.J. Young and P.A. Lovell, "Introduction to Polymers," ch. 4 <i>Polymer Structure</i> pp. 241-309 (crystallization of polymers, Tm, glass transition, etc.)		

# Surface vs. Bulk Hydrolysis: Göpferich's theory for polymer erosion<sup>1-4</sup>

Biodegradable solids may have differing *modes* of degradation:

Surface erosion – degradation from exterior only with little/no water penetration into bulk Bulk erosion – water penetrates entire structure and degrades entire device simultaneously



Fig. 1. Schematic illustration of the changes a polymer matrix undergoes during surface erosion and bulk erosion.

Polymers hydrolyzing by mechanisms II or III can be either surface or bulk eroding.<sup>5-7</sup>

Assuming that a polymer is water insoluble (initially) and that hydrolysis is the only mechanism of breakdown, the factors listed above all vary two rates of importance:

rate of water diffusion into polymer

rate of chain cleavage by water ions

The balance of these rates determines whether a polymer erodes from the surface in or by simultaneous degradation throughout the material:

Comparing velocities of water diffusion and chain cleavage:

Accounting for rate of water diffusion: Time required for water to diffuse a mean distance <x> into the solid polymer:

(1)  $t_{diff} = \langle x \rangle^2 \pi / 4D_{H2O}$ D<sub>H2O</sub> = effective diffusivity of water in polymer See Atkins Phys. Chem p. 770 for derivation

Random walk:



Fig. 10.18 One possible path of a random walk in three dimensions. In this general case, the step length is also a random variable.

(Atkins<sup>8</sup>)

Mean distance from origin traveled by water molecule after time t = <r> =  $(2D_{H2O}t)^{1/2}$ Mean distance traveled in x direction = <x> =  $2(D_{H2O}t_{diff}/\pi)^{1/2}$ EXPLAIN

Number of bonds in depth <x>:

(2)  $n = \langle x \rangle (bonds/cm^3)^{1/3} = \langle x \rangle (N_{Av}\rho/M_0)^{1/3}$   $N_{Av} = Avogadro's number$   $\rho = polymer density$  $M_0 = molecular weight of polymer repeat unit$ 

Accounting for rate of chain cleavage (k): probability that a bonds breaks in the interval (0,t):

(3)  $p(t) = ke^{-kt}$ 

where we have assumed that chain cleavage is a random event following Poisson kinetics

k = rate constant for bond hydrolysis

Therefore the mean lifetime of a single bond is given by:

(4) 
$$< t_c > = \prod_{0}^{d} p(t) dt = \prod_{0}^{d} e^{-kt} dt = \frac{1}{k} (kt+1) e^{-kt} \Big|_{0}^{d} = \frac{1}{k}$$

Time to degrade *n* bonds is a zero-order waiting time distributed according to a zero-order Erlang distribution:

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Mechanism (surface vs. bulk) is controlled by ratio of time for diffusion to time for hydrolysis, a dimensionless parameter analogous to a Deborah number:

#### Erosion number = $\varepsilon$

(5)  $\varepsilon \equiv t_{diff}/\langle t_c(n) \rangle = \langle x \rangle^2 k_c \pi / [4D_{H2O}\{\ln \langle x \rangle + (1/3)\ln (N_{Av}\rho/M_0)\}]$ 

• note <x> in denominator In should have same units as  $\rho$ , i.e. cm if  $\rho$  is in g/cm<sup>3</sup>

If <x> is replaced by the total thickness of a degrading sample, we can predict the mechanism of erosion:

- $\epsilon > 1$  bulk erosion
- $\epsilon = 1$  change in erosion mechanism
- $\epsilon < 1$  surface erosion



Fig. 2. Dependence of the erosion number,  $\varepsilon$ , on the diffusivity of water inside the polymer,  $D_{eff}$ , the dimensions of a polymer matrix, L, and the polymer bond reactivity,  $\lambda$ , calculated from equation 7. The white plane represents the area of surface erosion, the gray one the area of bulk erosion.

#### mass loss is linear for surface-eroding devices only

Table 1 Estimated values of  $\varepsilon$  and  $L_{\text{critical}}$  for selected degradable polymers

Chemical structure	Polymer	$\lambda$ (s <sup>-1</sup> )	E	$L_{\text{critical}}^{\text{b}}$	
	Poly(anhydrides)	$1.9 \times 10^{-3}$ Ref. [30]	11,515	75 µm	
	Poly(ketal)	$6.4 \times 10^{-5}$ Ref. [30]	387	0.4 mm	
	Poly(ortho esters)	$4.8 \times 10^{-5}$ Ref. [30]	291	0.6 mm	
	Poly(acetal)	$2.7 \times 10^{-8}$ Ref. [30]	0.16	2.4 cm	
$\begin{bmatrix} k \end{bmatrix}$ $- O - (CH_2)_5 - C +$	Poly(e-caprolactone)	$9.7 \times 10^{-8}$ Ref. [31]	0.1	1.3 cm	
[ ] [-,-,-,-,-,-]	$Poly(\alpha\text{-}hydroxy\text{-}esters)$	$6.6 \times \cdot 10^{-9}$ Ref. [30]	$4.0\times 10^{-2}$	7.4 cm	
[ ċн₃] [ ́н ́н o] - ́N-ċ-ċ	Poly(amides)	$2.6 \times 10^{-13}$ Ref. [30]	$1.5\times 10^{-6}$	13.4 m	

<sup>a</sup>For a 1 cm thick device,  $D = 10^{-8} \text{ cm}^2 \text{ s}^{-1}$  (estimated from Ref. [32]) and  $\ln \left[ \sqrt[3]{M_n/N_A(N-1)\rho} \right] = -16.5$ . <sup>b</sup> $D = 10^{-8} \text{ cm}^2 \text{ s}^{-1}$  (estimated from Ref. [32]) and  $\ln \left[ \sqrt[3]{M_n/N_A(N-1)\rho} \right] = -16.5$ .



Fig. 3. Critical thickness,  $L_{\text{critical}}$ , that a polymer device has to exceed to undergo surface erosion (calculated from Eq. (7), data shown in Table 1).

### Experimental demonstration of theory:

Transition of PLGA erosion from bulk to surface mode: degraded at basic pH (>12)- increased  $k_c$ , thus decreasing  $\epsilon$  << 1

Bulk (normal erosion at pH 7.4):



Fig. 4. Erosion profiles of poly( $\alpha$ -hydroxy esters) at pH 7.4: (a) PLA<sub>50</sub>11h ( $\bigcirc$ ) and PLA5017 ( $\Box$ ), (b) PLA<sub>25</sub>GA<sub>50</sub>8h ( $\blacklozenge$ ). PLA<sub>25</sub>. GA<sub>50</sub>14 ( $\blacklozenge$ ) and PLA<sub>25</sub>GA<sub>50</sub>47h ( $\blacksquare$ ).



(•) and PLA<sub>50</sub>17 ( $\Box$ ), PLA<sub>25</sub>GA<sub>50</sub>47h ( $\blacksquare$ ).

Surface (pH > 12):

SEM shown previously (Fig. 13) confirms transition to surface mode

# Synthesizing biodegradable macromolecules to tailor properties

# Approaches to molecular design

- Copolymerization
  - Control polymer hydrophobicity -> degradation rate
  - Control concentration of reactive groups
  - o Alter biocompatibility
    - What are the degradation products? Acidity/basicity? Toxicity? Biological effects?

• Vary Tm, Tg<sup>9</sup>, (mechanical properties)

# (SLIDE)

Table 3. Properties of poly( CL-co-DXO), poly(VL-co-DXO), and poly(LLA-co-DXO)				
Sample	% DXO in copolymer	Tg(DSC) [°C]	T <sub>m</sub> (DSC)["C]	
CD50	50	-56.8	27.8	
CD60	41	-57.8	27.2	
CD70	29	-55.5	36.0	
CD80	18	-61.0	42.8	
CD90	8	-65.6	50.5	
CD100	0	-65.9	57.6	
VD70	33	-56.7	28.0	
VD80	25	-56.1	37.7	
VD90	7	-59.9	46.0	
VD100	0	-63.4	57.5	
LD70	28	23.1	154.1	
LD85	13	41.1	178.8	
LD100	0	58.5	183.8	

Data adopted from [134]





wt.FRACTION LACTION FIGURE 2. Variation of glass transition temperature  $(T_g)$ of copolymers of caprolactone and DL-lactide as a function of DL-lactide content. Solid line is calculated relationship, based on Fox equation (from Reference 8).

Reactions on polymers/Polymer functionalization

## **Controlling Molecular Architecture**

We won't undertake an exhaustive description, but some of the important methods to be aware of:

- Condensation polymerization
  - Not very efficient, produces low molecular weight polymers (usually  $\leq$  10K g/mole)

$$\begin{array}{c} \mathsf{CH}_3 \mathsf{O} \\ \mathsf{HO}\text{-}\mathsf{CH}\text{-}\mathsf{C}\text{-}\mathsf{O}\mathsf{H} \end{array} \xrightarrow{\Delta} \begin{array}{c} \mathsf{CH}_3 \mathsf{O} \\ \mathsf{-}\mathsf{H}_2\mathsf{O} \end{array} \xrightarrow{\mathsf{CH}_3 \mathsf{O}} \\ \mathsf{-}(\mathsf{C}\mathsf{H}\text{-}\mathsf{C}\text{-}\mathsf{O})_{\mathsf{n}}^{-} \end{array}$$

- Has been found useful for growing dendritic polymers:
  - Prepared using AB<sub>2</sub>-type monomers

# (SLIDE)



Scheme 22. a) The divergent growth approach for the preparation of dendritic polyesters. b) The divergent approach for the preparation of dendrimer having 16-hydroxy groups

- Ring-opening polymerization
  - o Catalysis by stannous octoate (tin 2-ethyl hexanoate, FDA-approved)
    - Useful for polyesters (PLA, PCL, PGA, and their copolymers)<sup>10</sup>
    - Polymerization initiates from alcohol co-initiator groups by a coordination-insertion mechanism:



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example insertion:

# Proposed mechanisms: (on board)



For lactide and glycolide, each ring monomer opens to 2 lactic acid/glycolic acid moieties:



FIg.8. The chemical structure of glycolide and the resulting repeating unit

#### A variety of similar catalysts can be used to polymerize lactone ring monomers: (SLIDE)

β-propiolactone	°	β-PL
γ-butyrolactone	Č	γ-BL
β-butyrolactone	оСн_3	β-BL
δ-valerolactone	°	δ-VL
e-caprolactone	Č	ε-CL
1,5-dioxepan-2- one	Ç	DXO
R=H; glycolide	R	GA
R=CH <sub>3</sub> ; lactide	° ↓ R	LA

Table 1. Structure and designation of various lactones

Multi-alcohol initiators permit synthesis of multi-armed polymers:



- Living ring-opening polymerization
  - Coordination-insertion catalysts: e.g. aluminum isopropoxide<sup>10</sup>

## Provide control over molecular weight and MWD:







Allows the synthesis of block copolymers:



Hg.5. A schematic presentation of an ABA tri-block copolymer with two A-blocks (gray circles with dark centers) and one B-block (gray circles with light centers)

 Monomers polymerized sequentially, when block A is formed, monomer B is injected, etc. pendant peptide groups

Copolymerization of ring peptides with biodegradable monomers .

NaNO

**D-Alanine** 



e.g. Barrera et al<sup>12-14</sup>

- monomers must be synthesized from scratch 0
- bulky substituents make for highly inefficient ring-opening polymerization<sup>15</sup> 0

SOC

CO₂H

CHCI3, reflux NEt/ Pr2

Table 2. Effect of	the Concentration of 5 on	
Polymerizations Condu	acted at 136 °C for 48 h Using	t a
Catalyst to Mon	nomer Ratio of 1/1000 <sup>a</sup>	

mole % 5 <sup>b</sup>		vield			Ta	$T_{m}$
reaction	copolymer	(%)	$M_{\rm n}$	$M_{ m w}$	(°Ĉ)	(°Ĉ)
0.0	0.0	85	132 000	223 000	61.6	169.4
5.3	2.6	71	14 500	36 700	57.5	$155.3^{\circ}$
10.5	4.4	57	8 400	$23\ 000$	55.8	152.8 <sup>c</sup>
27.7	10.6	20	12 300	15 000	52.4	none
100.0		0				

<sup>a</sup> All the molecular weight data were obtained on protected copolymers. <sup>b</sup> Determined by <sup>1</sup>H NMR. Incorporated of 10 mol % 5 actually yields only 5 mol % lysine since each molecule of 5 contains one lysine residue and one lactic acid residue. <sup>c</sup> Indicates 2 or more melting endotherms.

- Network polymerization •
  - Photopolymerization of liquid precursors 0
    - E.g. polyanhydrides<sup>16,17</sup>
    - Allows formation of polymeric solids in situ from liquid precursors
      - Useful for dental restorations, bone fixation, tissue engineering
    - Curable through fiber optics or by shining light through tissue
    - UV or visible light initiators available



Fig. 1. Dimethacrylated anhydride monomers, methacrylated sebacic acid (MSA), methacrylated 1,3-bis(p-carboxyphenoxy) propane (MCPP) and methacrylated 1,6-bis(p-carboxyphenoxy) hexane (MCPH)), as well as a general polymerization and degradation scheme.





Fig. 3. Effect of light intensity on the rate of photopolymerization as a function of time for MSA polymerized with 0.5 wt% CQ and 0.5 wt% TEA.







cholesterol a vital component of cell membranes; stearic acid a natural fatty acid

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